

On the Luminescence of Ce^{3+} , Eu^{3+} , and Tb^{3+} in Novel Borate $\text{LiSr}_4(\text{BO}_3)_3$

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Abstract This paper reports on the photoluminescence (PL) and time-resolved properties of Ce^{3+} , Eu^{3+} , and Tb^{3+} in novel $\text{LiSr}_4(\text{BO}_3)_3$ powder phosphors. Ce^{3+} shows an emission band peaking at 420 nm under 350-nm UV excitation. Energy transfer from Ce^{3+} to Mn^{2+} takes place in the co-doped samples. Eu^{3+} shows red emission under near UV excitation. $\text{LiSr}_4(\text{BO}_3)_3:\text{Eu}^{3+}$ phosphor could be a suitable candidate for phosphor-converted solid state lighting. The luminescence lifetime is 2.13 ms for Eu^{3+} in $\text{LiSr}_4(\text{BO}_3)_3:0.001\text{Eu}^{3+}$. As Eu^{3+} concentration increasing, the decay curves deviate from exponential behavior. Tb^{3+} shows the strongest $^5\text{D}_4 \rightarrow ^7\text{F}_5$ emission line at 540 nm. Decay curves of $^5\text{D}_4 \rightarrow ^7\text{F}_5$ and $^5\text{D}_3 \rightarrow ^7\text{F}_5$ emission with different Tb^{3+} concentrations were also measured. Cross-relaxation process is discussed based on the decay curves.

Keywords Phosphor · Luminescence · Decay curve · Borate

Introduction

Rare earths and transition metal ion doped phosphors have attracted more attention due to their extensive applications in the fields of lighting, display and medical science as well

as fundamental studies [1–6]. Phosphors are usually made from a suitable host material, to which an activator is added. The well known host materials are oxides [7], nitrides and oxynitrides [1, 8, 9], sulfides, phosphates [10], halides [11], silicates [12, 13] and borates. Among the borates based phosphors, $\text{InBO}_3:\text{Eu}$, $\text{InBO}_3:\text{Tb}$, $\text{Sr}_6\text{P}_5\text{BO}_{20}:\text{Eu}$, $\text{SrB}_4\text{O}_7:\text{Eu}$ and $\text{SrFB}_2\text{O}_3:\text{Eu(II)}$ have been utilized in industry [14]. Recently, borates based phosphors have been studied due to their lower synthesis temperature and other advantages [15–21]. The compound, $\text{MM}'_4(\text{BO}_3)_3$ ($\text{M} = \text{Li}, \text{Na}, \text{K}$; $\text{M}' = \text{Ca}, \text{Sr}, \text{Ba}$), represents a new structure type [22, 23]. Jiang and his co-workers have reported the thermoluminescence characteristics of $\text{NaSr}_4(\text{BO}_3)_3:\text{Ce}^{3+}$ [24], $\text{LiSr}_4(\text{BO}_3)_3:\text{Ce}^{3+}$ [25], and $\text{KSr}_4(\text{BO}_3)_3:\text{Ce}^{3+}$ [26] phosphors. In the former papers we reported on the photoluminescence properties of $\text{NaCa}_4(\text{BO}_3)_3:\text{Eu}^{3+}$ and $\text{NaSr}_4(\text{BO}_3)_3:\text{Ce}^{3+}$, Mn^{2+} [27, 28]. These materials emit intense visible light and are promising phosphors for practical application. Among these compounds, $\text{LiSr}_4(\text{BO}_3)_3$ crystallizes in the cubic space group $Ia\bar{3}d$ with large lattice parameters: $a = 14.95066(5)$ Å. In this paper, the luminescence and time-resolved properties of Ce^{3+} , Eu^{3+} , and Tb^{3+} in $\text{LiSr}_4(\text{BO}_3)_3$ are described in detail.

Experimental

Samples of $\text{LiSr}_4(\text{BO}_3)_3:\text{RE}$ ($\text{RE} = \text{Ce}^{3+}$, Eu^{3+} , Tb^{3+}) were prepared according to the standard solid-state technique. High-purity starting materials Li_2CO_3 (Aldrich, 99.9%), SrCO_3 (Aldrich, 99.9%), H_3BO_3 (Aldrich, 99.9%, 10 mol% excess to compensate the evaporation in the heating processes), Eu_2O_3 (Aldrich, 99.99%), Tb_4O_7 (Aldrich, 99.99%), and CeO_2 (Aldrich, 99.99%) were used. The well mixed materials were pre-sintered at 500°C for 3 h, then

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annealed at 800°C for 12 h in air (for Eu^{3+} doped samples) or weak CO reducing atmosphere (for Tb^{3+} , Ce^{3+} doped samples) with an intermediate grinding. Li_2CO_3 was added as a charge compensator. The structural characteristics of $\text{LiSr}_4(\text{BO}_3)_3:\text{RE}$ samples were checked by X-ray diffraction (XRD) patterns using a Rigaku D/max 2200 Diffractometer with $\text{Cu K}\alpha$ radiation at 40 kV, 30 mA. All the samples are single phase. The PL emission and excitation spectra were measured by Fluorolog-3 Fluorescence Spectrophotometer with 450 W Xenon lamps. The luminescence decays were measured by monitoring the given emission from the samples under 266 nm pulsed laser excitation. Decay profiles were recorded by the 500 MHz digital oscilloscope (LeCroy 9350A) in which the signal was fed from PMT.

Results and Discussion

The PL excitation and emission spectra of $\text{LiSr}_4(\text{BO}_3)_3:0.01\text{Ce}^{3+}$ are given in Fig. 1. Either 410 or 460 nm emission are monitored, the excitation spectra are similar to each other and consist of two distinct broad-bands peaking at about 280 and 340 nm. The absorption bands are assigned to the crystal field levels of the $4f/5d$ states for Ce^{3+} , and the lowest $5d$ -level absorption for Ce^{3+} in $\text{LiSr}_4(\text{BO}_3)_3$ is about 3.65 eV (340 nm). Under 350-nm UV excitation PL emission spectrum shows an asymmetry emission band peaking at 420 nm. The luminescence is ascribed to a $5d-4f$ transition on Ce^{3+} , corresponding to the transitions of $5d$ excited state to ${}^2\text{F}_{5/2}$ and ${}^2\text{F}_{7/2}$ ground states.

It is well known that the transitions of Mn^{2+} are spin and parity forbidden; the emission and excitation intensities are often weak under UV light excitation. Ce^{3+} can absorb UV light, especially the near-UV in some matrixes. Many authors have reported that Mn^{2+} emission can be strengthened by means of energy transfer from Ce^{3+} to Mn^{2+} [3, 29–31]. In these cases, a significant spectral

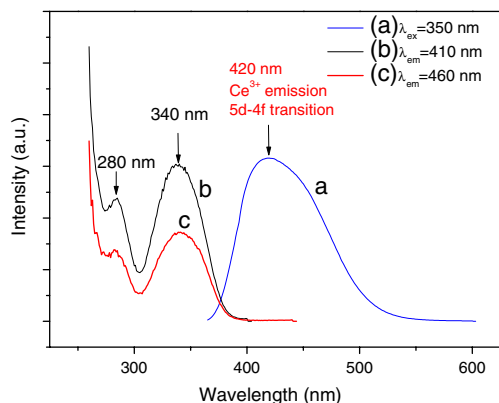


Fig. 1 Emission (a) and excitation (b and c) spectra of the luminescence of $\text{LiSr}_4(\text{BO}_3)_3:0.01\text{Ce}^{3+}$

overlap between the emission peak of Ce^{3+} and the excitation peak of Mn^{2+} (${}^6\text{A}_1 \rightarrow {}^4\text{A}_1$, ${}^4\text{E}$ transition) has been observed, indicating that efficient resonance-type energy transfer from Ce^{3+} to Mn^{2+} takes place. Based on the Ce^{3+} emission in $\text{LiSr}_4(\text{BO}_3)_3$ matrix, we also studied the possibility of $\text{Ce}^{3+} \rightarrow \text{Mn}^{2+}$ energy transfer in Ce^{3+} , Mn^{2+} co-doped phosphors. The PL excitation and emission spectra of $\text{LiSr}_4(\text{BO}_3)_3:0.04\text{Ce}^{3+}, 0.16\text{Mn}^{2+}$ sample are shown in Fig. 2. It can be seen that the emission spectrum exhibits a strong blue band (420 nm) and a weak red band (600 nm). The two bands are attributed to the $d \rightarrow f$ transition of Ce^{3+} and ${}^4\text{T}_1-{}^6\text{A}_1$ transition of Mn^{2+} , respectively. For the Mn^{2+} single dopant sample, the Mn^{2+} emission is too weak to be observed because of forbidden transition. So the red band emission of Mn^{2+} ions in co-doped sample should be attributed to the energy transfer from Ce^{3+} to Mn^{2+} . Unfortunately, the transfer efficiency is not satisfactory. The excitation spectra of the 420 and 600 nm emission correspond to the $f \rightarrow d$ transitions in the Ce^{3+} ion, which also confirms the energy transfer taking place between Ce^{3+} and Mn^{2+} ions.

Eu^{3+} doped $\text{LiSr}_4(\text{BO}_3)_3$ phosphors show red emission under UV excitation. Figure 3 shows the PL excitation and emission spectra of $\text{LiSr}_4(\text{BO}_3)_3:1.0\text{Eu}^{3+}$. Under 260 and 393 nm excitation, the Eu^{3+} emission consists of the well-known transitions from the ${}^5\text{D}_0$ to ${}^7\text{F}_J$ levels. The emission peaks observed at 578, (585, 595), 613, 650, and (685, 700) nm are assigned to the transitions ${}^5\text{D}_0 \rightarrow {}^7\text{F}_0$, ${}^5\text{D}_0 \rightarrow {}^7\text{F}_1$, ${}^5\text{D}_0 \rightarrow {}^7\text{F}_2$, ${}^5\text{D}_0 \rightarrow {}^7\text{F}_3$, and ${}^5\text{D}_0 \rightarrow {}^7\text{F}_4$, respectively. Moreover, the red light emission does not quench even if the x value is up to 1.0 in $\text{LiSr}_4(\text{BO}_3)_3:x\text{Eu}^{3+}$ system. The excitation spectrum ($\lambda_{\text{em}}=613$ nm) consists of broad excitation band peaking at about 260 nm and some intense $f \rightarrow f$ absorption lines. The excitation band should be assigned to the charge transfer transition from O^{2-} to Eu^{3+} in the host lattice. The absorption of BO_3 groups may be situated at higher energy level [32, 33]. The sharp lines in the 300–550 nm range are intraconfigurational $4f-4f$ transition of Eu^{3+} ions in the host

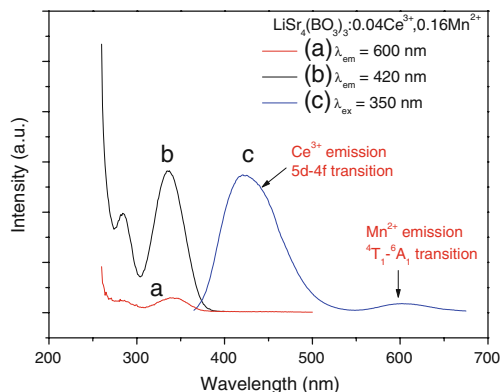


Fig. 2 Excitation (a and b) and emission (c) spectra of the luminescence of $\text{LiSr}_4(\text{BO}_3)_3:0.04\text{Ce}^{3+}, 0.16\text{Mn}^{2+}$

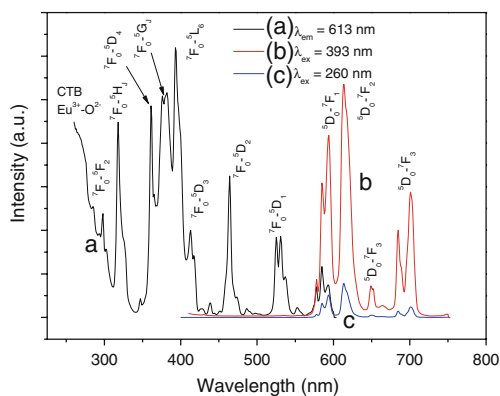


Fig. 3 Excitation (a) and emission (b) spectra of the luminescence of $\text{LiSr}_4(\text{BO}_3)_3:1.0\text{Eu}^{3+}$

lattices. It is noted that the phosphors exhibit intense absorption around 400 nm, and this absorption band match the emission of near UV InGaN chip; so $\text{LiSr}_4(\text{BO}_3)_3:\text{Eu}^{3+}$ phosphor could be a suitable candidate for phosphor-converted solid state lighting.

When the luminescent centers have different local environments, the associated ions relax at different rates [34, 35]. The decay curves of ${}^5\text{D}_0 \rightarrow {}^7\text{F}_2$ emission for $\text{LiSr}_4(\text{BO}_3)_3$ doped with different Eu^{3+} concentrations ($0.001 \leq x \leq 1.0$) under excitation at 266 nm are shown in Fig. 4. It can be seen that the decay curves are single exponential when $x \leq 0.04$. The luminescence lifetime derived from the fitted curves are 2.13 ms for Eu^{3+} in $\text{LiSr}_4(\text{BO}_3)_3:0.001\text{Eu}^{3+}$. As Eu^{3+} concentration increases, the decay curves deviate from exponential behavior, and the non-exponential change becomes more prominent with increasing Eu^{3+} content, which indicates that more than one relaxation process exists. With the Eu^{3+} concentration increasing, the distances between Eu^{3+} ions shorten; then the probability of an energy transfer among Eu^{3+} ions

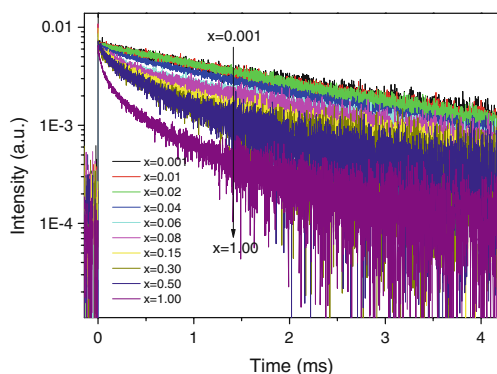


Fig. 4 Decay curves of ${}^5\text{D}_0 \rightarrow {}^7\text{F}_2$ emission for $\text{LiSr}_4(\text{BO}_3)_3$ doped with different Eu^{3+} concentrations under excitation at 266 nm. The signals were detected at 616 nm

becomes more frequent. The mechanism of energy transfer is under investigation.

The PL emission and excitation spectra of $\text{LiSr}_4(\text{BO}_3)_3:0.08\text{Tb}^{3+}$ phosphor are shown in Fig. 5. In the excitation spectrum of the ${}^5\text{D}_4 \rightarrow {}^7\text{F}_5$ transition (540 nm), the sharp absorption lines at 317, 351, 367, 377, and 485 nm are due to the transitions from ${}^7\text{F}_6$ to ${}^5\text{D}_3$, ${}^5\text{G}_3$, ${}^5\text{L}_{10}$, ${}^5\text{D}_3$, and ${}^5\text{D}_4$, respectively. The absorption band peaking at 280 nm can be ascribed to the $4f^8 \rightarrow 4f^7 5d^1$ transition of Tb^{3+} ion. Tb^{3+} ion with $4f^8$ configuration has complicated energy levels, so luminescence spectrum consisting of many lines due to ${}^5\text{D}_3 \rightarrow {}^7\text{F}_j$ is observed. In this case, the emission spectrum shows the strongest ${}^5\text{D}_4 \rightarrow {}^7\text{F}_5$ emission line at approximately 540 nm. Moreover, the ${}^5\text{D}_3$ emission is weaker than that of ${}^5\text{D}_4$ emission even in the sample with very low Tb^{3+} concentration ($\text{LiSr}_4(\text{BO}_3)_3:0.001\text{Tb}^{3+}$), and it does not disappear even with higher Tb^{3+} concentration ($\text{LiSr}_4(\text{BO}_3)_3:0.20\text{Tb}^{3+}$).

The decay curves of ${}^5\text{D}_3 \rightarrow {}^7\text{F}_5$ emission (415 nm) for $\text{LiSr}_4(\text{BO}_3)_3$ doped with different Tb^{3+} concentrations under excitation at 266 nm are shown in Fig. 6. A double exponential decay behavior of the luminescence emission has been observed. All the decay curves can fit very well to a double exponential function. The short lifetime is about 1 μs and decreases with increasing Tb^{3+} concentration. The fast decay component may be attributed to the host lattice emission or some kind of Tb^{3+} cluster or pair [36]. The slow decay component can be ascribed to the ${}^5\text{D}_3 \rightarrow {}^7\text{F}_5$ transition of Tb^{3+} . There is no obvious change to the lifetime of this transition, and the fitted lifetime is about 40 μs .

Figure 7 shows the decay curves of ${}^5\text{D}_4 \rightarrow {}^7\text{F}_5$ emission (540 nm) for $\text{LiSr}_4(\text{BO}_3)_3$ doped with different Tb^{3+} concentrations under excitation at 266 nm. It can be seen that the decay curves show a nearly single exponential decay for those samples with lower Tb^{3+} concentration

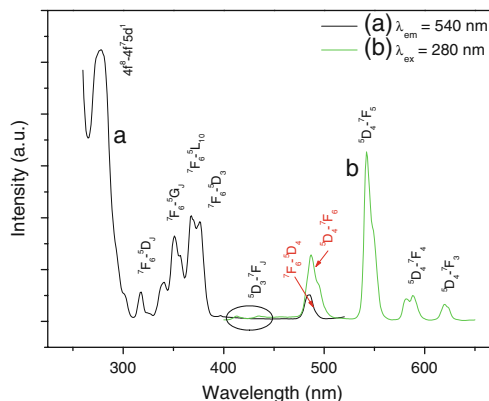


Fig. 5 Excitation (a) and emission (b) spectra of the luminescence of $\text{LiSr}_4(\text{BO}_3)_3:0.08\text{Tb}^{3+}$

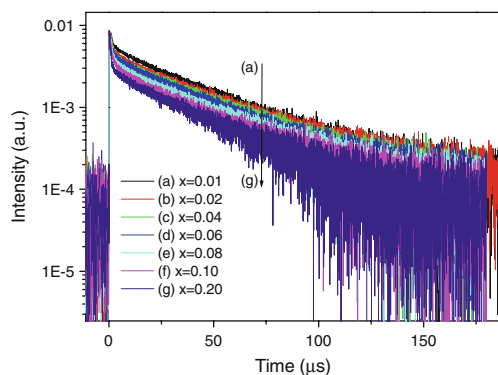


Fig. 6 Decay curves of ${}^5D_3 \rightarrow {}^7F_5$ emission for $\text{LiSr}_4(\text{BO}_3)_3$ doped with different Tb^{3+} concentrations under excitation at 266 nm. The signals were detected at 415 nm

(e.g., $x=0.01$ and 0.02). A single exponential fit derives a decay time about 3.6 ms. It is also noted that the decay curves (curves a and b) exhibit an initial rise phenomenon (marked with black circle in Fig. 7). We attempted to fit these curves with double exponential equation [37, 38]:

$$I(t) = I_0 \left[\exp\left(\frac{-t}{\tau_1}\right) + A \exp\left(\frac{-t}{\tau_2}\right) \right] \quad (1)$$

Where, I and I_0 is the luminescence intensity, t is the time, τ_1 is a decay time, τ_2 is a decay and/or rise time, A is the ratio of the two parts of the function. When $A < 0$, the second term represents rising, and for $A > 0$, the second term represents a further decay. The results of the fitting indicate that the best agreement between experimental data and theoretical fits is obtained, and the result of $\text{LiSr}_4(\text{BO}_3)_3:0.01 \text{ Tb}^{3+}$ sample is shown in Fig. 8. The values of parameters are $\tau_1 = 3.76$ ms (decay time), $\tau_2 = 60 \mu\text{s}$

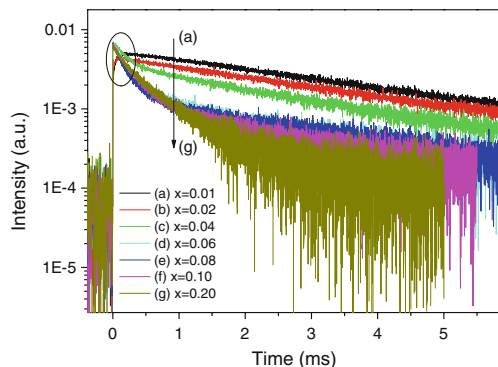


Fig. 7 Decay curves of ${}^5D_4 \rightarrow {}^7F_5$ emission for $\text{LiSr}_4(\text{BO}_3)_3$ doped with different Tb^{3+} concentrations under excitation at 266 nm. The signals were detected at 540 nm

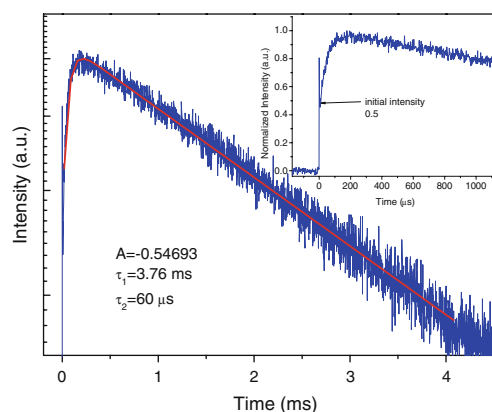


Fig. 8 Simulated decay curve of using equation (1) for $\text{LiSr}_4(\text{BO}_3)_3:0.01 \text{ Tb}^{3+}$ phosphor (the inset shows the initial intensity of ${}^5D_4 \rightarrow {}^7F_5$ transition)

(rise time) and $A = -0.54693$. The rise time of $60 \mu\text{s}$ is close to the value of the decay time of the 5D_3 emission in this matrix, indicating the initial slow rise is attributed to the ${}^5D_3 \rightarrow {}^5D_4$ cross-relaxation process. The initial intensity of 0.5 represents about 50% of 5D_4 population feeding directly from the $5d$ level and 50% population feeding from higher energy level (5D_3) of Tb^{3+} by cross-relaxation process [39]. For $x \geq 0.04$, the decay curves become nonexponential due to the radiationless process involving Tb^{3+} ions [36], and the mechanism is under investigation. The initial rise part disappears quickly, which may be due to the higher Tb^{3+} concentration [36].

Conclusions

In summary, we have reported on the synthesis and PL properties of Ce^{3+} , Eu^{3+} , and Tb^{3+} in novel $\text{LiSr}_4(\text{BO}_3)_3$ powder phosphors. The $\text{LiSr}_4(\text{BO}_3)_3:\text{Ce}^{3+}$ phosphor shows a blue emission peaking at 420 nm under excitation by 350 nm UV light. The emission is due to the $5d-4f$ transition of Ce^{3+} . In the Ce^{3+} , Mn^{2+} co-doped samples energy transfer from Ce^{3+} to Mn^{2+} occurs. The $\text{LiSr}_4(\text{BO}_3)_3:\text{Eu}^{3+}$ phosphor shows red emission under near UV excitation. $\text{LiSr}_4(\text{BO}_3)_3:\text{Eu}^{3+}$ phosphor could be a potential candidate for phosphor-converted solid state lighting. The decay curves of $\text{LiSr}_4(\text{BO}_3)_3:x\text{Eu}^{3+}$ phosphors are single exponential when $x \leq 0.04$, and deviate from exponential behavior, with increasing Eu^{3+} content, which indicates that more than one relaxation process exists. The $\text{LiSr}_4(\text{BO}_3)_3:\text{Tb}^{3+}$ phosphor shows strongest ${}^5D_4 \rightarrow {}^7F_5$ emission line at 540 nm. The decay curves of ${}^5D_3 \rightarrow {}^7F_5$ emission can be fitted well by double exponential function. The decay curves of ${}^5D_4 \rightarrow {}^7F_5$ emission exhibit a nearly single exponential decay for those samples with dilute Tb^{3+}

concentration with lifetime of 3.6 ms, and they exhibit an initial rise phenomenon, which is attributed to the $^5D_3 \rightarrow ^5D_4$ cross-relaxation process.

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